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The Structure and Mechanical Characteristics of the Hypereutectic Silumin (Al–22–24 wt.% Si), Irradiated by a Pulsed Electron Beam

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Abstract—Irradiation ("SOLO" setup, High-Current Electronics Institute SB RAS) of hypereutectic Al- (22–24) Si composition silumin samples with an intense pulsed electron beam was carried out. The formation of a multiphase submicron nanocrystalline structure is revealed. It was found that after modification, the hardness of silumin surface layer increases by more than 2 times. Tensile tests were carried out for flat proportional silumin samples in the initial and after irradiation states. An increase in the ductility of irradiated samples was demonstrated.

Keywords—intense pulsed electron beam, aluminum, hypereutectic silumin, silicon, structure

I. INTRODUCTION

Silumin is a silicon and aluminum alloy. Silumin is an interstitial solution. The silicon content varies from 4 to 29 wt.%. At higher silicon values, the alloy loses its ductility. The ultimate solubility of silicon in aluminum is 1.65% at the eutectic temperature, and is 0.05% at the room temperature. In turn, aluminum in silicon dissolves only in an amount of 0.5% [1, 2]. According to the diagram, eutectic, hypoeutectic and hypereutectic solutions can be in equilibrium [3]. According to the percentage of silicon, silumins are divided into hypoeutectic (Si < 11%), eutectic (Si = 11–12.5%), hypereutectic (Si > 12.5%) [1].

Hypereutectic silumin is a promising aluminum alloy. Due to such properties as good fluidity, low weight, corrosion resistance can be used for the manufacture of machine parts and mechanisms. The structure of this alloy consists of eutectic and primary grains of silicon, as well as intermetallic compounds. These alloys belong to natural composites. Maria Rygina Institute of High Current Electronics SB RAS 2/3 Academichesky Ave., 634055, Tomsk, Russia National Research Tomsk Polytechnic University 30 Lenina Ave., 634050, Tomsk, Russia L-7755me@mail.ru

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Depending on the silicon content, the hardness of the hypereutectic silumin changes, the higher the silicon content, the higher the hardness [4]. Silicon is a fairly brittle material, with an increase in its concentration in the Al-Si alloy, processes of embrittlement of silumin occur. Primary silicon grains can reach sizes of 100 microns or more, what has a significant impact on the ability to operate the material. A significant effect on the size of the primary silicon grains is exerted by hydrogen, which gets from the air when casting the product into the mold. It is known that saturation of the melt with hydrogen complicates the further diffusion of silicon. At the moment, there are technologies to eliminate the fragility of silumin by alloving. The main disadvantage of alloving is that different modifiers are used to grind each phase. For example, beryllium binds iron [5], the addition of finely dispersed titanium or antimony to the melt reduces the size of primary silicon by 4 and 2 times, respectively [6]; to increase the dispersion of primary crystals of the α -phase of silumins, Al – B - Ti and Al - Zr ligatures are used. Aluminum-silicon eutectics are the most effectively modified by sodiumcontaining fluxes. Primary silicon crystals in hypereutectic silumins are mainly ground up by phosphorus-containing ligatures. In this case, the action time of sodium-containing fluxes usually does not exceed 0.5 h [7].

Another way to modify the structure of silumin is to change the method of casting products from this alloy. Thus, using centrifugal casting with water cooling, it was possible to obtain a microdimensional structure [8]. Silumins do not form chemical compounds, therefore, do not lend themselves to heat hardening treatment. The size of silumin grains and, accordingly, the strength characteristics of the material can be varied by changing the cooling rate [9].

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Of particular difficulty in casting products from silumins by traditional methods is the effect of heeredity, which manifests itself in maintaining the grain size when using finished castings as a charging material [10].

The method of modifying silumin by irradiating the surface of a part with an intense pulsed electron beam of micro- and submillisecond exposure duration allows one to achieve high heating and cooling rates. This method, as shown by studies performed on samples of hypoeutectic and eutectic silumins [3, 11-15], is one of the promising methods for modifying the surface layer of an alloy with a thickness of up to 1000 µm.

The aim of this work is to study the structure and mechanical properties of hypereutectic silumin after modification by a pulsed electron beam of submillisecond duration of exposure.

II. EXPERIMENTAL SETUP

The material under study was hypereutectic silumin, the silicon concentration was (22-24) wt. %. Silumin of hypereutectic composition was prepared in a shaft-type laboratory electric resistance furnace with silicon carbide heaters in a painted stainless steel crucible. As a charge, we used commercially pure aluminum A7 (GOST 11069-2001) and silicon Kr0 (GOST 2169-69). The castings obtained were rectangular plates with dimensions of 55×120×20 mm. From the obtained ingot, rectangular specimens with dimensions of 15×15×5 mm were cut. Flat silumin specimens were prepared in the form of double-sided blades for tensile tests in accordance with GOST 1497-84 [16]. Some of the samples were irradiated with an intense pulsed electron beam ("SOLO" setup [17]) before testing. Irradiation mode: accelerated electron energy 18 keV, electron beam energy density 30 J / cm^2 , pulse repetition rate 0.3 s⁻¹, electron beam exposure duration 150 µs, number of irradiation pulses 5. The irradiation mode was selected according to thermal calculations [18]. The samples were examined by optical (µVizo-MET-221), scanning (SEM-515 Philips), and transmission (JEM 2100 F) electron microscopy. Hardness tests were carried out on a PMT-3 installation. Tensile fracture tests were performed on an Instron 3369 setup.

RESULTS AND DISCUSSION III.

A. Initial Structure of Hypereutectic Silumin

The initial structure of silumin specimens is represented by eutectic grains, primary silicon grains, and inclusions of intermetallic compounds based on silicon, aluminum, and iron. In fig. 1 shows an image of the initial structure of hypereutectic silumin. The primary silicon grains size reaches 100 microns. The study of the energy spectra of the surface of the original silumin, carried out by X-ray microanalysis, showed the presence of 23.0 wt. % silicon, the rest aluminum, traces of iron were also revealed ($\leq 1 \text{ wt.}\%$).

B. Structure of Modified Silumin Subjected to Pulsed Electron **Beam Treatment**

The surface layer of hypereutectic silumin was modified by irradiating the samples with an intense pulsed electron beam (30 J/cm2, 150 μ s, 0.3 s⁻¹, 5 pulses). As shown in [18], the selected mode allows melting the silumin surface layer up to 60 um thick. In this case, the silicon grains and intermetallics inclusions dissolution in the aluminum melt is recorded. Cellular structure is formed in this layer as a result of highspeed crystallization. The cells sizes vary within 250-400 nm (Fig. 2).



Fig. 1. Optical image of the initial structure of hypereutectic silumin.



⊐ 2.0 µm

Fig. 2. Electron microscopic image of the hypereutectic silumin structure irradiated with an electron beam.

Nanosized particles of silicon and intermetallic compounds are located along the cell boundaries.

Fig. 3, a demonstrates the absence of primary grains, as well as gas pores that are formed in silumin at the casting stage.

The silicon content is 22.0 wt. %, iron -1.0 wt.%, the rest is aluminum.

A high-speed crystallization characteristic of hypereutectic silumin, initiated by an irradiation with a pulsed electron beam, is the formation of globular silicon grains in the surface layer. In fig. 4 shows images of the silumin structure obtained in the characteristic X-ray radiation of silicon (fig. 4, a) and aluminum (fig. 4, b) atoms. Analyzing the results presented in this figure, it can be noted that the silicon grain size varies from 0.8 μ m to 1.5 μ m. In addition, silicon in the form of thin (up to 100 nm thick) interlayers is located, as noted above, along the high-speed crystallization of aluminum cells interfaces (Fig. 4, a). It should be noted that a change in the shape of silicon inclusions from predominantly lamellar in the initial state to globular, after irradiation with a pulsed electron beam, will facilitate the plasticization of silumin, reducing the likelihood of its brittle destruction due to cracking in silicon plates.





Fig. 3. SEM image (a) and energy spectra (b) of the silumin surface after modification with an electron beam (30 J/cm², 150 μ s, 0.3 s⁻¹, 5 pulses).

C. Mechanical Properties of Modified Silumin

The microhardness of the hypereutectic silumin samples in the cast state was 1600 MPa; after irradiation the microhardness of silumin increased in 2.4 times and amounted to 3840 MPa. Tensile tests of hypereutectic silumin samples in the cast and electron-beam-irradiated states showed that the tensile strain of the cast silumin was 3.6%. After irradiation with a pulsed electron beam, the tensile strain increased by 1.5 and amounted to 5.4%.



Fig. 4. Image of hypereutectic silumin structure irradiated with a pulsed electron beam (30 J/cm², 150 μ s, 0.3 s⁻¹, 5 pulses), obtained in the characteristic X-ray radiation of silicon (a) and aluminum (b) atoms. Silumin layer at a distance of 20 microns from the irradiation surface.

IV. CONCLUSION

The irradiation of silumin with an intense pulsed electron beam in the mode of melting the surface layer (30 J/cm², 150 μ s, 0.3 s⁻¹, 5 pulses) was carried out. It was found that high-speed crystallization leads to the formation of aluminum

submicro-nanocrystalline cellular type structure. Interlayers enriched with silicon and iron atoms were revealed at the cells boundaries. It is shown that high-speed crystallization leads to the formation of globular silicon grains. It was found that irradiation of the silumin surface with a pulsed electron beam leads to an increase in hardness by more than two times and deformation in tension by a factor of 1.5.

Thus, the modification method proposed in this work, based on the use of an intense pulsed electron beam, can significantly increase the strength and plastic properties of hypereutectic silumin.

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