



Article Surface Topography Control of TA2 Pure Titanium in Laser Shock Peening

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Abstract: Laser shock peening (LSP) induces an irregular topography on the treated metal surface, thereby reducing the gain effect of the metal fatigue property caused by compressive residual stress. A technique named laser shock imprinting (LSI) is proposed in this paper to guide plastic deformation on a titanium surface. An FEM simulation and experiment were conducted to explore the embossment forming process and residual stress distribution of TA2 pure titanium. The simulated results show that the embossment on the sample surface went through five stages, namely, static, growth, rebound, fluctuation and stabilization, under a single LSI. With an increase in loading pressure, the contact pressure between the sample and contact foil increased along with increasing embossment height. Sufficient loading pressure could induce a difference in residual stress between the zones with and without embossment. The experimental results show that the height and clarity of embossment increased with increasing laser energy, a result similar to that of the simulation. In addition, compared with LSP, the sample treated by LSI had a lower surface roughness and flatter surface profile.

Keywords: laser shock imprinting; embossment forming process; residual stress distribution; laser energy; surface profile

1. Introduction

Laser shock peening (LSP) is an effective technology to improve the high-cycle fatigue performance of aero-engine blades [1,2]. Various dislocation structures and high-magnitude compressive residual stress are introduced into the material surface, improving the yield strength, micro-hardness, wear resistance, corrosion resistance and fatigue property of the material [3–5]. However, the plastic deformation induced by the laser shock wave produces a spot-like pit structure. Severe plastic deformation may limit the propagation of residual stress on the material surface and damage the uniform distribution of residual stress. The residual stress accumulates gradually in the center of the pit and a residual stress hole is formed, which leads to a decrease in the fatigue property of metals [6–8]. Therefore, it is important to minimize the adverse effects of plastic deformation on the material surface while ensuring the strengthening effect of LSP.

Many studies have been conducted about surface roughness control in LSP. Dai et al. [9] found that the surface roughness (Ra) of LY2 aluminum alloy treated by LSP was stable at 0.58 μ m when polyester tape was used as the absorbing layer. Pardhu et al. [10] found that the surface roughness of a metal subjected to LSP with adhesive tape as the absorbing layer was lower than that with black paint as the absorbing layer. Salimianrizi et al. [11] found that the surface roughness of the material could be reduced by changing the way of overlapping and rate of spots. A square light spot was used by Zou et al. [12,13] to treat the surface of a compressor blade, which reduced the surface roughness of the blade



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Copyright: © 2022 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). and improved the processing efficiency. However, due to the scattering effect of the laser shock wave passing through the absorbing layer, an irregular interlaced zone still existed on the surface of the material treated by the square spot. Dai et al. [14] covered the sample surface with a high-strength metal foil in the LSP process, and they found that the use of high-strength foil could weaken the overlapping effect of the adjacent spot to a certain extent and reduce the surface roughness, but without an entire elimination.

Laser shock imprinting (LSI) is proposed in this paper to guide the surface topography of TA2 pure titanium. The method was as follows: a contact foil with high hardness and 0.1 mm thickness was added between the absorbing layer and the titanium. A regular texture was fabricated on the contact foil. A schematic of LSI is shown in Figure 1a. When a high-magnitude shock wave induced by a laser beam was applied to the contact foil, the plastic deformation on the titanium sample was guided directly under the interaction between the contact foil texture and the titanium surface. An FEM simulation and experiment were conducted to observe the surface topography control of LSI, where the FEM simulation focused on the embossment forming process and the FEM experiment focused on the embossment profile. The residual stress distribution is also studied in this paper. This work aims to study the surface topography deformation behavior during LSI and to provide a reference for controlling the surface topography in LSP.



Figure 1. LSI schematic (a) and finite element model (b).

2. FEM Simulation and Experiment

2.1. FEM Simulation

A two-dimensional model was used to simulate the LSI process, as shown in Figure 1b. The upward movement of the contact foil would be limited by the aluminum foil in the experiment. Therefore, in order to make the moving process of the contact foil similar to that in the experiment, 0.1 mm aluminum was placed on the contact foil to act as a barrier. The shock pressure was directly loaded onto the contact foil surface.

The mesh division of the FEM model was as follows: Firstly, the surface deformations of the barrier and contact foil were not considered in the research process; therefore, these two parts were divided into coarse meshes, where the mesh size was 0.1 mm \times 0.1 mm and the mesh types were CPS4R (two-dimensional, four nodes) and CPS3 (two dimensions), respectively. Secondly, the surface layer with a depth of 0.3 mm was subdivided into meshes with a unit size of 0.002 mm \times 0.002 mm. The middle layer adopted transition meshes (unit size of 0.02 mm \times 0.02 mm), and the bottom layer adopted coarse meshes (unit size of 0.2 mm \times 0.2 mm). The mesh types were CPS4R and CPS3, and the total number of elements was 273,350.

Due to the fact that the embossment and sink in the special zone of the sample treated by LSI belong to deformation at the ultra-high strain rate, the effect of strain on the material properties should be considered [15]. The Johnson–Cook model of isotropic hardening was adopted in this paper [16], and the mechanical parameters are shown in Table 1.

Table 1. Mechanical parameters of 65 manganese steel and TA2 titanium alloy.

Material	<i>P</i> (kg/m ³)	A (MPa)	B (MPa)	n	С	E (GPa)	v
65 Mn steel	7810	980	2000	0.83	0.0026	200	0.288
Pure titanium	4500	430	1000	0.3	0.047	108	0.3

The energy of the laser shock wave was equivalent to the uniform pressure in an ultra-short time. Berthe et al. [17] found that the action time of laser shock wave to material with a water confining layer was about 2–3 times longer than that without a confining layer. In this paper, the pulse duration of the nanosecond laser was 20 ns, so the duration of loading pressure was 40–60 ns. The time evolution of pressure conformed to the triangular ramp curve mode [18], which increased from zero to peak in a period of 20 ns, and then decreased from peak to zero in the next 40 ns. The laser spot was a circle with a diameter of 3 mm; therefore, the pressure loading area diameter was 1.5 mm, taking the left axis as the central axis because a two-dimensional symmetrical model was used in this simulation. According to the formulation derived from Fabbro et al. [19], the parameters of LSI were laser energies of 4 J, 5 J and 6 J, corresponding to loading peak pressures of 3 GPa, 4 GPa and 5 GPa, respectively.

The dynamic calculation was divided into three analysis steps: In the first step, the ABAQUS/Explicit code was adopted, the time increment was 5 ns, and the total time length was 500 ns, which made the embossment on the material surface form entirely. The time increment of the second step was 250 ns and the total time length was 15 μ s, which made the material surface stable. In the third step, static and general modes were used to balance the residual stress distribution of the material. The total time length of this step was 0.5 s.

2.2. Experiment

The chemical composition of TA2 pure titanium is shown in Table 2. TA2 pure titanium was cut into a rectangular block with a size of $10 \text{ mm} \times 10 \text{ mm} \times 2 \text{ mm}$, and then ground and polished with sandpaper and a polishing belt, respectively. Finally, the samples were ultrasonically cleaned with alcohol.

Table 2. Chemical composition of TA2 pure titanium.

Elements	Fe	С	Ν	Н	0	Ti
Weight%	≤ 0.30	≤ 0.10	≤ 0.05	≤ 0.015	≤ 0.25	balance

The contact foil in this paper was 65 Mn steel with a thickness of 0.1 mm, where the hardness was 500 HV and the tensile strength was 735 MPa. The chemical composition of 65 Mn steel is shown in Table 3. The 65 Mn steel contact foil was cut to a size of 10 mm \times 10 mm \times 0.1 mm. A groove-like array texture was fabricated with a nanosecond pulsed laser, and then the grooved surface was polished. Figure 2 presents the schematic of the contact foil size (a) and the texture distribution (b), where the length, width and depth of the groove were 1 mm, 0.1 mm and 0.05 mm, respectively.

Table 3. Chemical composition of 65 Mn steel.

Elements	С	Si	Mn	S	Cr	Ni	Cu	Fe
Weight%	0.62-0.70	0.17-0.37	0.9–1.2	≤ 0.035	≤ 0.25	≤ 0.30	≤ 0.25	balance



Figure 2. Schematics of contact foil size (a) and texture size (b).

An Nd-YAG: GAIA nanosecond pulsed laser (Thales, Paris, French) was used to conduct LSI and LSP experiments, as shown in Figure 3. The processing parameters were as follows: the laser pulse width was 20 ns, the pulse energy was 4–6 J, the wavelength was 1064 nm, the treated area was a single laser spot, the laser pulse frequency was 1 Hz and the impact time was 1. The diameter of the laser spot reaching the target was about 3 mm. The sample was fixed onto the workbench with double-sided tape. Black polyester tape with a thickness of 200 μ m was used as the absorbing layer. In LSI, the textured side of the contact foil was closely attached to the polished surface of the sample, but contact foil was not used in LSP. Running deionized water with a thickness of 1–1.5 mm was used as the confining layer.



Figure 3. GAIA nanosecond pulse laser shock system: (**a**) laser system, (**b**) computer-integrated control system, (**c**) workbench and (**d**) path control system.

A VK-X200 laser confocal microscope (Keyence, Osaka, Japan) was used to observe the imprinted topography on the sample surface.

3. Results and Discussion

3.1. FEM Simulation

Figure 4 presents the two-dimensional profile of the embossment on the sample surface treated with different loading peak pressures. It can be seen that the top of the embossment is flat, but with different shapes on the bottom. The bottoms of the embossments corresponding to peak pressures of 3 GPa and 4 GPa are relatively flat, while that corresponding to the pressure of 5 GPa shows a slightly bilateral asymmetry. This indicates that the value of peak pressure affects the forming inclination of the embossment bottoms. When the peak pressures are 3 GPa, 4 GPa and 5 GPa, the heights of the embossment are 1 μ m, 1.7 μ m and 2.7 μ m, respectively.



Figure 4. Profile of the embossments at different peak pressures.

The height of the embossment is set as the displacement between the highest point and the lowest point of the embossment. Figure 5 shows the dynamic height variation of the embossment within 15,000 ns. The change in embossment forming height within the first 500 ns is also shown in Figure 5b to clearly observe the dynamic deformation behavior of the embossment. The main forming stage of the embossment is within the first 200 ns. After 200 ns, the height of the embossment only fluctuates slightly, which has a small effect on the final determination of the embossment height. It can be seen from Figure 5a that the forming process of the embossment can be divided into five stages, that is, static, growth, rebound, fluctuation and stabilization. In general, the final height of the embossment increases with the increase in peak pressure. The forming process of the embossment can be summarized specifically as follows. When pressure acts on the contact foil, it first propagates inside the foil for 10 ns, and then arrives at the sample surface. The zone of the sample corresponding to the texture zone of the contact foil is embossed, and the embossment is initiated. The embossment increases to the maximum height and then rebounds due to the attenuation of shock pressure. After rebounding to a certain extent, the residual energy of the shock wave makes the height of embossment show a fluctuating effect. The final embossment is generated after a longer stabilization stage.



Figure 5. Evolution of embossment height at different peak pressures with time, where (**b**) shows in detail the first 500 ns of (**a**).

Contact pressure plays the most important role in the formation of embossment. Figure 6 presents the contact pressure evolution of the embossment bottom at different peak pressures. The contact between the foil and sample surface mainly occurs in the first 200 ns. When the peak pressure increases from 3 GPa, to 4 GPa, to 5 GPa, the contact pressure increases from 1800 MPa, to 2300 MPa, to 3000 MPa, respectively, and the action time of loading pressure is also extended from 120 ns, to 150 ns, to 150 ns, respectively. The contact pressure evolution of the first 50 ns is extracted. It can be seen that it takes 10 ns for the pressure wave to propagate to the sample surface.



Figure 6. Contact pressure evolution of the embossment bottom at different peak pressures.

Figure 7 presents the velocity evolution of the top and the bottom of the embossment at different peak pressures. It can be seen that the bottom starts to move at 10 ns, which is the same as the initial moment of loading pressure acting on the sample surface. The top starts to move at the moment of 35 ns, which is later than the moment of the bottom

moving. In addition, compared with the velocity evolution of the top, the bottom has higher values of velocity and larger fluctuation. The velocities of the two parts approach zero at the moment of 250 ns, and then keep a small fluctuation. To sum up, the pressure wave propagates to the sample surface at the moment of 10 ns, and then the bottom undergoes downward deformation movement upon the impact of the contact foil. Due to the guidance of the groove texture on the foil surface, the bottom begins to squeeze the top, which makes the top deform upward at the moment of 35 ns. The main period of the velocities of the top and the bottom is 0–250 ns, followed by a slow fluctuation stage. The displacement movement of the bottom and top together promotes the formation of the embossment.



Figure 7. Velocity evolution of the embossment top (a) and bottom (b) at different peak pressures.

Figure 8 presents the residual stress distribution inside the sample at different peak pressures. It can be seen that compressive residual stress with high amplitude appears in the sample, whose value increases with the increase in peak pressure. Magnified images of the special embossment zone and the surrounding zone are shown in Figure 8 to carefully observe the residual stress distribution in these two zones. The residual stress distribution in the embossment and surrounding zones treated with a loading peak pressure of 3 GPa is uniform, with values between -150 MPa and -200 MPa. The residual stress distribution in the embossment and surrounding zones treated with a peak pressure of 4 GPa is nonuniform. There is an obvious residual stress difference in these two zones, where the compressive residual stresses of the embossment and surrounding zones are approximately -150 MPa and -250 MPa, respectively. The residual stress difference can also be observed in the embossment surrounding zones of the sample treated with 5 GPa peak pressure, and an extremely high tensile residual stress exists at the edge of the embossment. This is due to the stress concentration caused by the special surface topography [20]. To sum up, a difference in residual stress appears on the embossment and surrounding zones of the sample surface treated with the peak pressures of 4 GPa and 5 GPa, which is attributed to the different plastic strain levels led by the different deformation and contact modes in the corresponding zones of the sample. There is no obvious residual stress difference between the embossment and surrounding zones in the sample treated with the 3 GPa peak pressure, which indicates that sufficient loading pressure is required to induce the stress difference between these two zones.



Figure 8. Residual stress distributions inside the sample and around the specific embossment at different peak pressures: (**a**) 3 GPa, (**b**) 4 GPa and (**c**) 5 GPa.

The residual stress distributions on the sample surface and at a 20 μ m depth below the surface at a peak pressure of 5 GPa are shown in Figure 9. Tensile residual stress is found at the edge of the embossment. Compressive residual stress is dominant in the range of 0–1.4 mm, tensile residual stress is dominant at 1.4–1.8 mm, and there is almost no residual stress at 1.8–3 mm. Regarding the 20 μ m depth below the surface, the residual stress distribution is still consistent with the spatial distribution of the surface topography, but it is more moderate as a whole. The compressive residual stress at 0–1.4 mm is between –180 and –350 MPa, which shows the low compressive residual stress in the central zone of the embossment. The compressive residual stress at 1.4 mm starts to weaken and gradually transforms into tensile residual stress, and finally decreases to zero at 2 mm.



Figure 9. Residual stress distribution on the sample surface (**a**) and residual stress distribution at a depth of 20 μ m below the surface (**b**), under a peak pressure of 5 GPa.

3.2. Experiment Results

Figure 10 shows the embossments on the sample surface treated with different laser energies. It can be seen that there was not a large deformation on the surface of the 4 J sample, with the average embossment height being 3.5 μ m. As for the 5 J sample, a certain tilt appeared on the sample surface, and the spatial distribution of embossments became nonuniform. The average embossment height was 5 μ m. When the laser energy increased

to 6 J, the sample surface tilt increased further; meanwhile, the clarity of the embossments also increased, accompanied by the average embossment height increasing to 6 μ m. To sum up, with an increase in laser energy, the tilt on the sample surface increases, and the height of embossments also increases. The results of the experiment are similar to that of the FEM simulation.



Figure 10. Surface topography of the samples treated with different laser energies.

According to the above analysis, the dynamic forming process of the embossment at a single shock pressure can be divided into five stages, as shown in Figure 11 (taking the sample treated with a peak pressure of 5 GPa as an example).



Figure 11. Schematic of the embossment forming process on the sample surface subjected to LSI treatment.

In the first stage (static stage, 0–10 ns), the loading pressure wave propagates from the contact foil to the sample surface and drives the interaction between the foil and the sample. This stage is only the propagation of the pressure wave, without the generation of contact pressure on the sample surface.

In the second stage (embossment growth stage, 10–120 ns), the material surface starts to respond after contact pressure is generated on the sample. When the accumulated stress exceeds the dynamic yield strength of the material, deformation occurs on the sample surface and the extrusion happens at the moment of 35 ns under the guidance of the contact foil. The embossment is formed and reaches the maximum height at 120 ns.

In the third stage (embossment rebound stage, 120–200 ns), the pressure wave attenuates over time, which leads to a decrease in the deformation rate of the sample surface. The elastically deformed part of the sample starts to rebound, and the velocities of the compression and extrusion at the sample top and bottom decrease. This makes the embossment height decrease in this stage.

In the fourth stage (embossment fluctuation stage, 200–15,000 ns), the internal rebound of the sample and the gradual attenuation of the pressure wave mutually restrict each other, and jointly promote height fluctuation of the embossment. In this stage, there is a small variation in the formation height of the embossment.

In the fifth stage (stabilization stage, 15 μ s–0.5 s), the embossment on the sample surface has been formed generally in the first 15 μ s. In this stage, the surface topography is further stabilized. The residual stress is regularly distributed in the embossment and surrounding zones. The height of the embossment reduces slightly under the effect of residual stress until the final balanced state.

Figure 12 presents the surface morphology of (a) as-received, (b) 5 J LSP and (c) 5 J LSI samples, and the profiles of all samples. It can be seen that the original surface roughness of the as-received sample was 1.904 μ m, and there was only low tilt in the surface profile. Regarding the 5 J LSP sample, the surface roughness increased to 4.273 μ m and large fluctuation occurred on the surface. As for the 5 J LSI sample, the surface roughness decreased to 2.259 μ m, which was 47% lower than that of the 5 J LSP sample. Only low tilt appeared on the 5 J LSI sample, whose surface profile was flatter than that of the 5 J LSP sample. This indicates that the irregular morphology induced by the laser shock wave on the material surface can be eliminated by contact foil. Compared with LSP, the roughness induced by LSI on the material surface is lower, with a value close to that of the as-received sample.



Figure 12. Surface morphology of (a) as-received, (b) 5 J LSP and (c) 5 J LSI samples, and (d) profiles.

4. Conclusions

TA2 pure titanium was treated by LSI technology. The surface response under an ultra-high shock wave was simulated by FEM, which involved the surface deformation process, contact pressure change and residual stress distribution in the sample. The FEM simulation was combined with experiments to understand the forming process of the embossment on the titanium surface. Finally, the surface morphologies of titanium treated by LSP and LSI were compared to verify the guidance effect of LSI on the sample surface. The conclusions are as follows:

- (1) The height of the embossment is proportional to the loading peak pressure. With an increase in loading pressure, the contact pressure and time between the sample surface and the contact foil increase, and the height of the embossment also increases. The forming process of the embossment on the sample surface can be divided into the stages of static, growth, rebound, fluctuation and stabilization.
- (2) A difference in residual stress between the embossment and surrounding zones is generated when the loading pressure increases to a certain extent, and extremely high tensile residual stress also appears on the edge of the embossment.
- (3) With an increase in laser energy, the clarity and height of the embossment increased, and the tilt of the sample surface also increased.
- (4) Compared with the LSP sample, the surface roughness of the sample treated by LSI was 47% lower, with a flatter surface profile. The irregular morphology induced by the laser shock wave on the sample surface was changed into a single tilt under the effect of the contact foil.

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